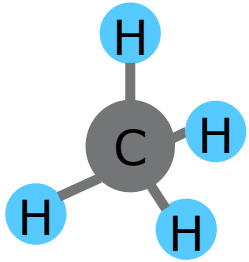


Hydrogen and Integrity of Existing Natural Gas Networks

HIPS-NET workshop

Alfons Krom





Natural gas / methane and hydrogen are similar gases.

But there are differences.

For integrity of network hydrogen molecules
can be source of hydrogen atoms.

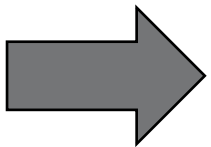
Hydrogen atoms may cause hydrogen embrittlement of the steel pipe.

- The interaction of hydrogen atoms and steel may have a negative effect on the mechanical behaviour of the steel. The general term for this degrading effect is *hydrogen embrittlement*.
- Hydrogen embrittlement is well known in the petrochemical industry and in the welding world.
- The form of HE depends on many variables: one of them is the hydrogen atom concentration in the steel

Hydrogen concentration in steel

Hydrogen gas loading, no oxide layer present

- Chemical equation: $\text{H}_2 \rightleftharpoons 2[\text{H}]_{\text{Fe}}$
- Sieverts' law: $C_{\text{H}} = k\sqrt{p_{\text{H}_2}}$
- k: solubility depends on steel and temperature
- Hydrogen pressure 81 bar / 8,1 MPa (a)



0,25 atomic parts per million = 1 H-atom per 4 million Fe-atoms

Comparison with other hydrogen sources *wet H₂S gas, welding, active cathodic protection*

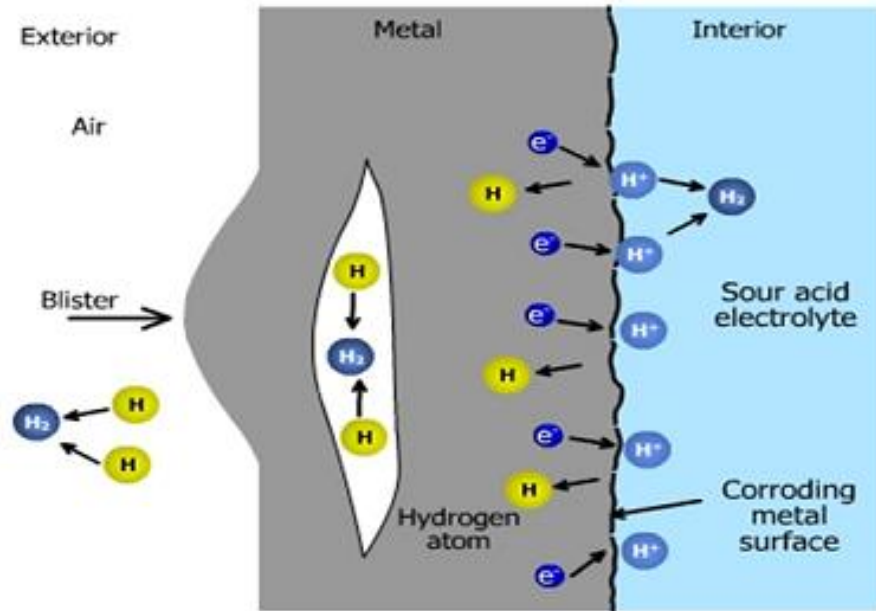
no oxide
layer
present

H source	H concentration [atomic ppm]	equivalent pressure [bar (a)]
81 bar H ₂	0,25	81
wet 0,01 bar H ₂ S ^a	14	7062
active cathodic protection ^b	56	11438
3 ml H ₂ /100 g weld electrode	150	14519

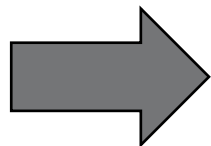
a) K. van Gelder et al., Hydrogen-induced cracking: determination of maximum allowed H₂S partial pressures, Corrosion, vol 42, no 1, 36-43 1986

b) D.X. He et al., Effect of cathodic potential on hydrogen content in a pipeline steel exposed to NS4 near-neutral pH soil solution, Corrosion, 778-786 2004

wet H₂S gas -> Hydrogen-induced cracking recombination of H-atoms in defects in the steel wall

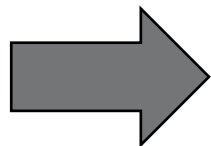
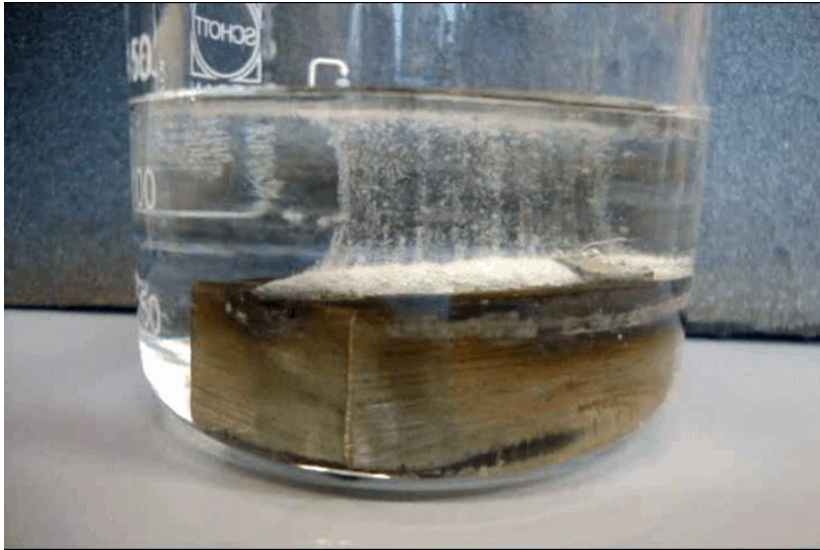


blister



Pipelines: no HIC because hydrogen gas pressure is significantly lower.

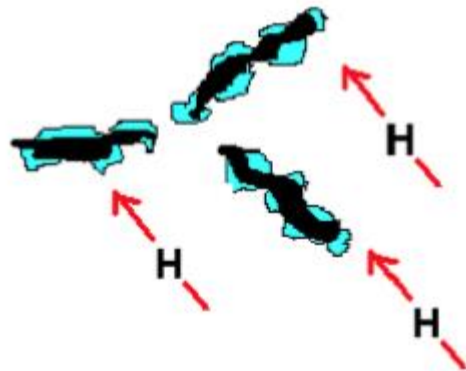
Hydrogen from the weld electrode -> cold cracking



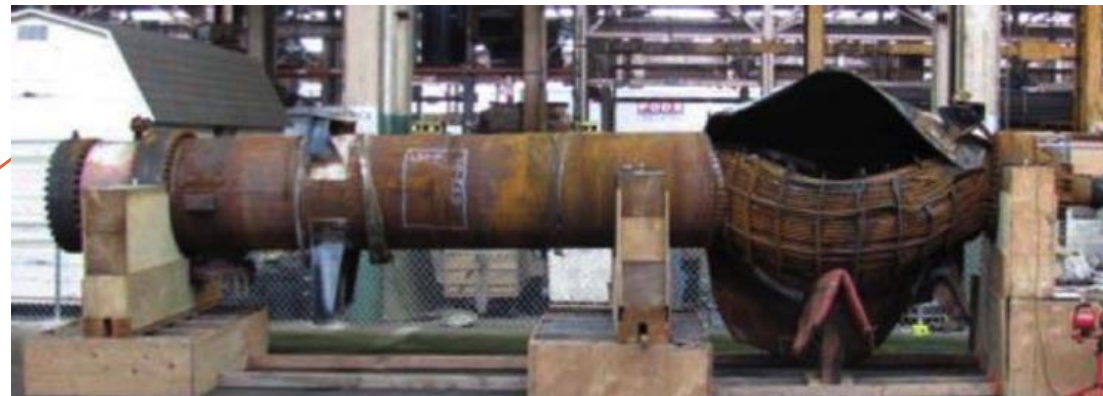
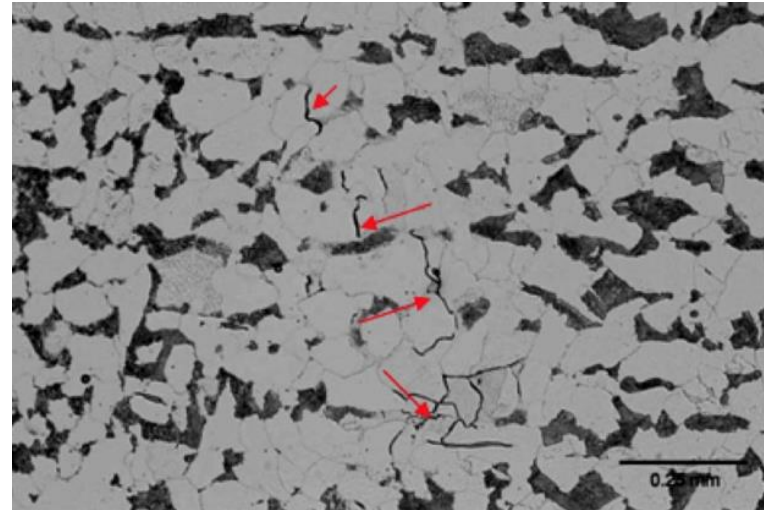
Pipelines: no cold cracking because hydrogen concentration is too low.

Hydrogen reacts with carbon -> Hydrogen attack

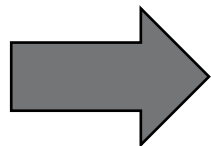
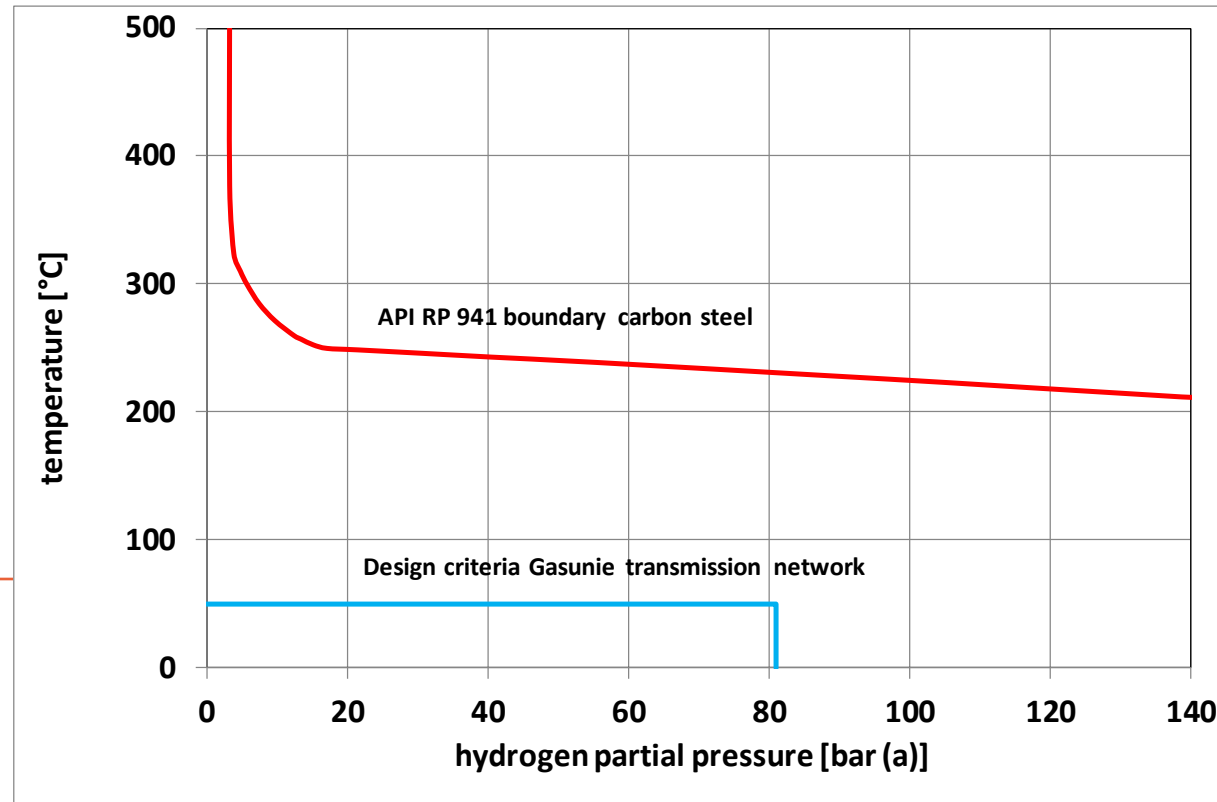
CH₄ gas pockets (blue)



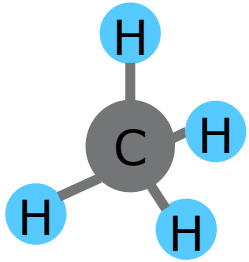
H is driven into the steel by heat & pressure, and reacts with the Fe₃C to form CH₄ gas



Hydrogen attack depends on temperature and pressure

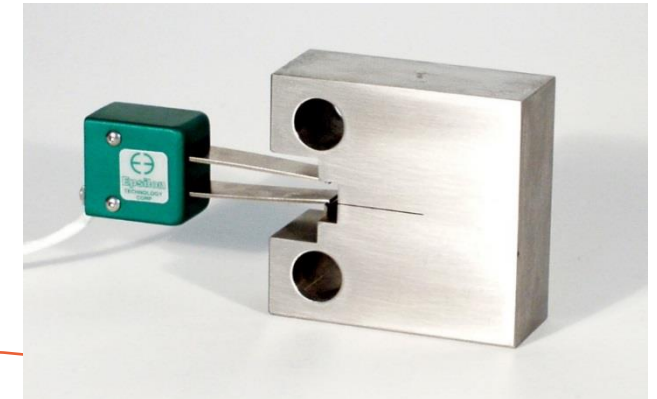
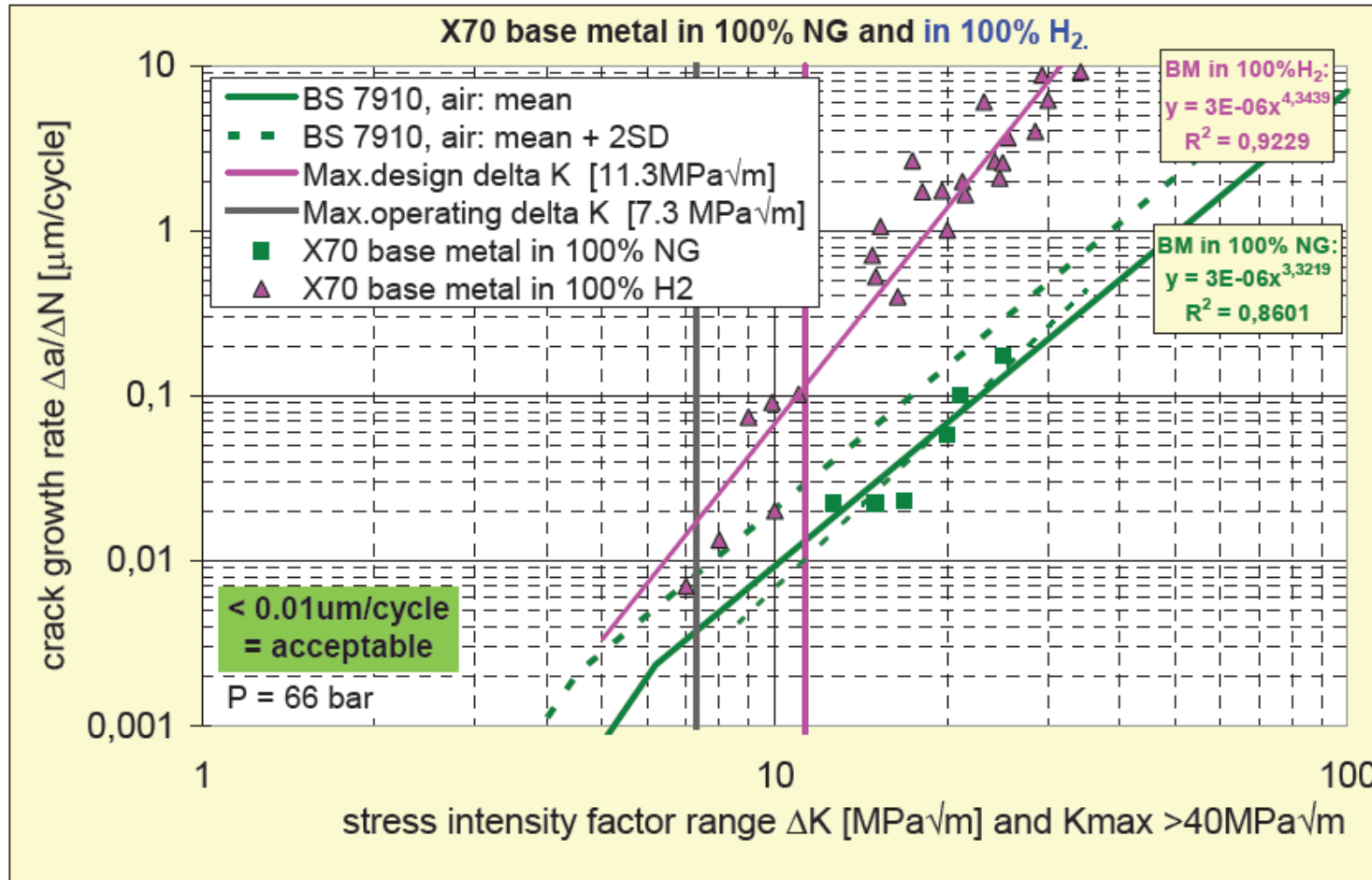


Pipelines: no hydrogen attack because pressure and temperature is significantly lower.



When does hydrogen gas have a negative effect
on steel behaviour?

Fatigue crack growth in H₂ (purple) and in CH₄/air (green)

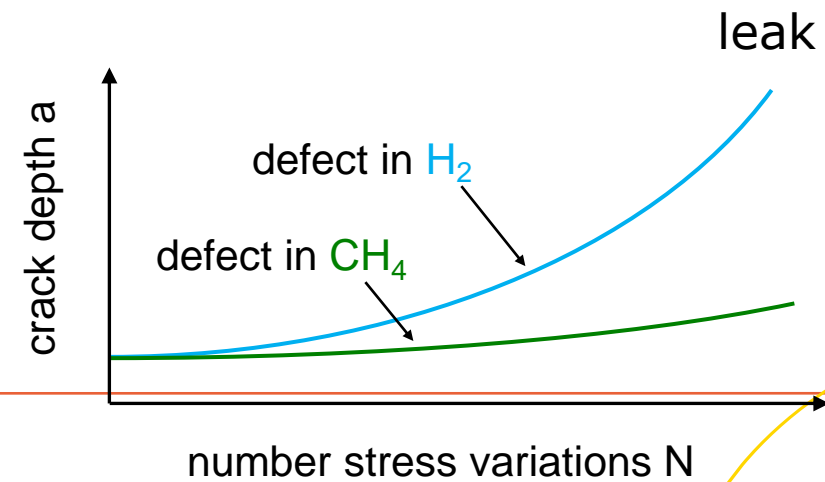
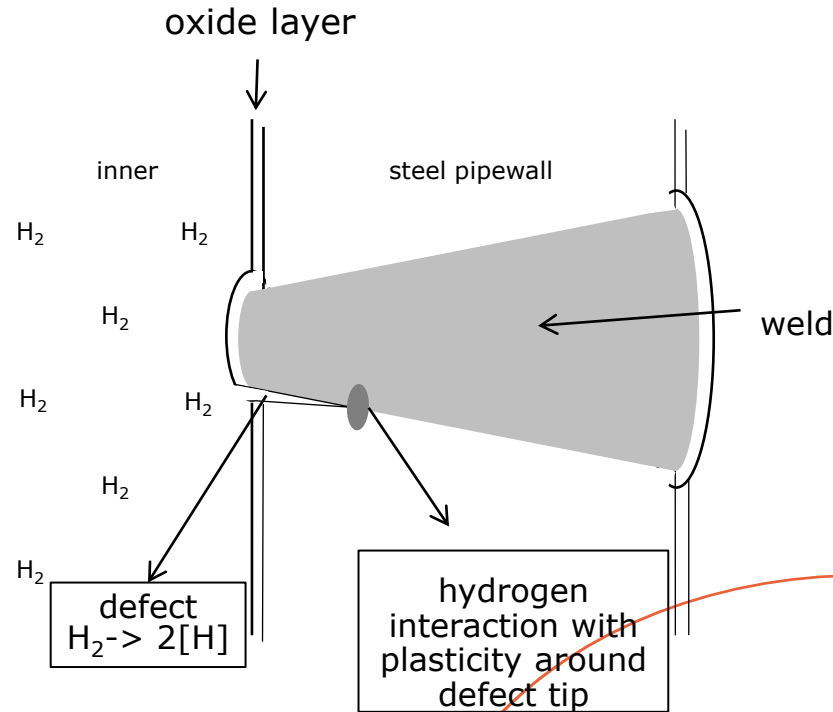


compact tension specimen

crack driving force $\Delta K = \Delta\sigma\sqrt{\pi a}$

0,01 $\mu\text{m/cycle}$, 100 year 1 cycle per day = 0,37 mm crack growth in air

Scenario for hydrogen-enhanced fatigue crack growth



girth weld of pipeline with inner weld defect

The approach for dealing with hydrogen-enhanced fatigue of existing natural gas pipelines

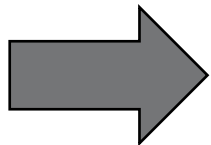
	description	example
1	choice of a assumed crack-like defect in the pipeline	weld defect typically: 3 mm high and 50 mm long
2	the stress intensity factor ΔK of the assumed defect ("crack force")	see next slide
3	the fatigue load (stress range and number of cycles)	daily pressure variation of 10% of internal pressure
4	the required lifetime of the pipeline	100 year
5	the fatigue crack growth rate at ΔK of the assumed defect in H ₂ gas	see next slide
6	the lifetime of the assumed defect	see next slide

Step 2 & 3 crack force ΔK and stress range $\Delta\sigma$
assumed defect is 3 by 50 mm in longitudinal
pipe weld or girth weld in 14,1 mm, 48", 66 bar

pressure cycle		crack orientation in weld	stress [MPa]	stress range $\Delta\sigma$ [MPa]	crack force variation ΔK [MPa \sqrt{m}]
[%]	[bar]				
10	6,6	longitudinal	292	29	3,4
		girth	150	15	1,7

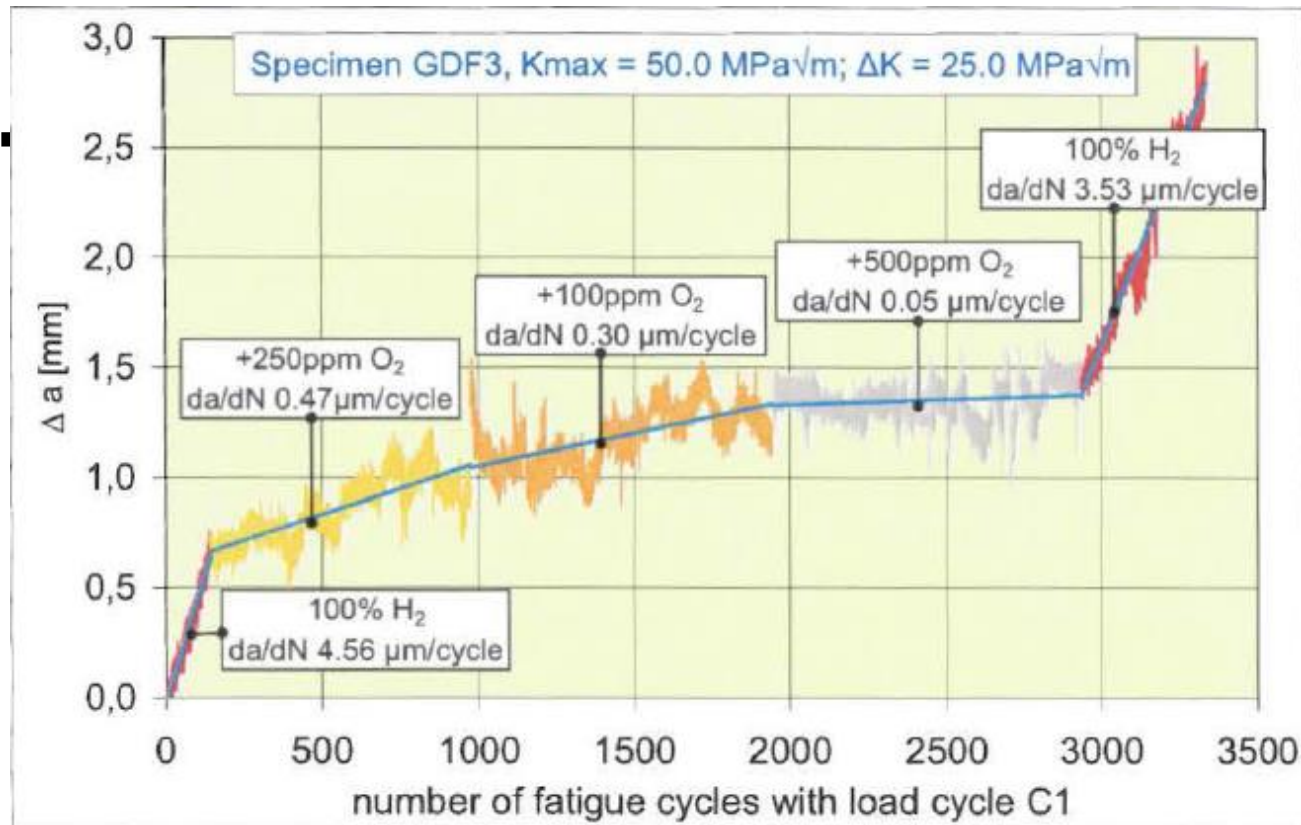
Step 5 & 6 ΔK of the assumed defect in H₂ gas and the crack growth over 100 year

pressure cycle [bar]	crack orientation in weld	crack force variation ΔK [MPa \sqrt{m}]	crack growth rate [mm/cycle]	crack growth 100 year [mm]
6,6	longitudinal	3,4	$6,1 \cdot 10^{-7}$	0,022
6,6	girth	1,7	$3,0 \cdot 10^{-8}$	0,001



Crack growth is so small over a period of 100 year, 100% H₂ at 66 bar does not impose an integrity risk.

Fatigue crack growth in the presence with O₂



frequency $0,00164 \text{ s}^{-1}$, 66 bar H₂, steel X52=L360

Conclusion (1/3)

Where hydrogen gas is being transported in pipelines at ambient temperatures and moderate pressures, the relevant hydrogen degradation mechanism is hydrogen-enhanced fatigue crack growth. When taking this degradation mechanism into account, 100% hydrogen gas up to the design pressure can be transported in existing natural gas pipelines without affecting the integrity of the pipeline during its lifetime.

But

Conclusion (2/3)

Though the integrity may not be affected by the hydrogen, it does not mean that hydrogen can actually be transported in the existing pipeline. Hydrogen is a smaller molecule than the methane molecule and the ignition energy is much lower.

Conclusion (3/3)

So before hydrogen can be transported in an existing pipeline the following has to be considered:

- is the leak tightness of existing valves (internal and external) sufficient?
- is the leak tightness of existing flanges sufficient?
- do the risk contours of the pipeline become larger because the risk assessment for hydrogen is different?
- can operational and maintenance activities be performed in a safe manner?
- is welding on a live pipeline possible?

Nothing new ...

“The major technical problem with transmission of hydrogen gas at high pressure is the possibility of slow fatigue crack growth from existing cracks or crack-like defects in the pipe body or weld.”

E. Anderson et al. Geneva Research Centre in “Analysis of the potential transmission of hydrogen by pipeline in Switzerland”

Proceedings of the 2nd World Hydrogen Energy Conference, Zurich, Switzerland, 21-24 August **1978**